



Mosaic and gradient single crystals for gamma ray Laue lenses

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Gamma Wave 2005

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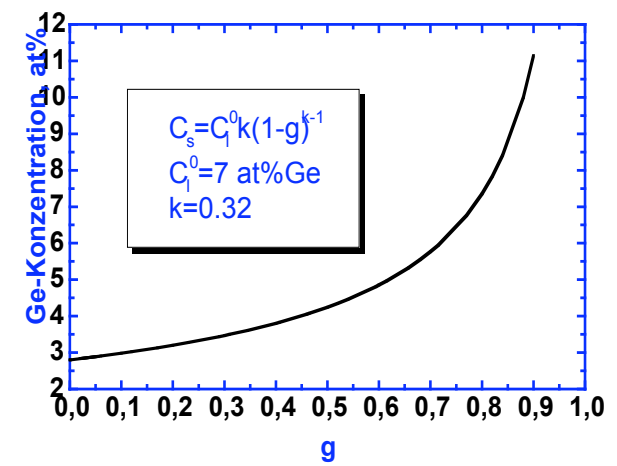
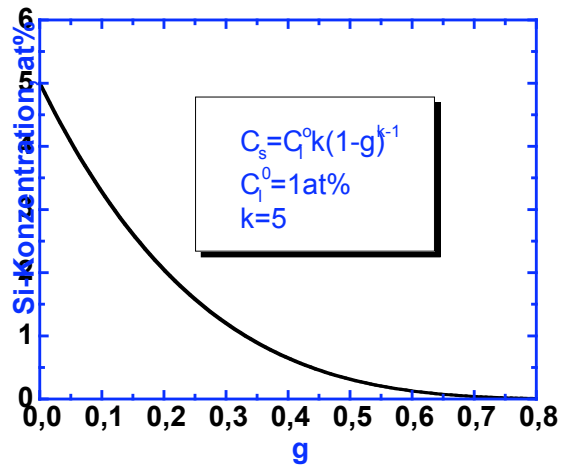
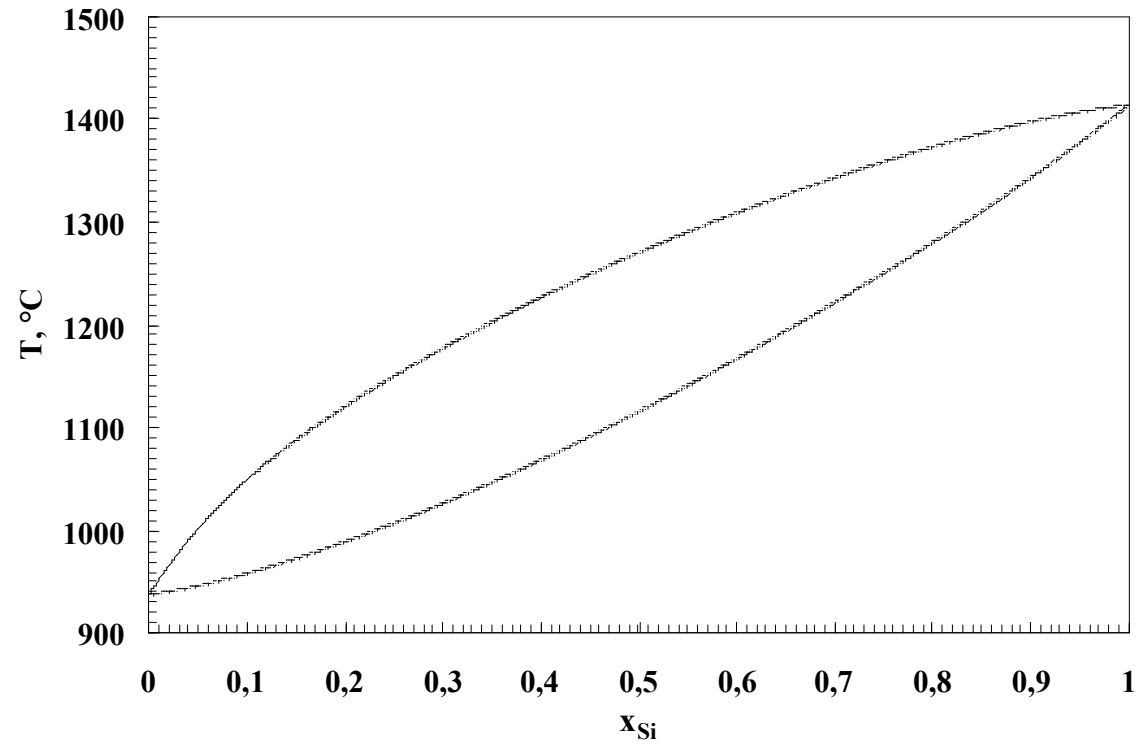
Espace St. Jacques, Bonifacio, Corsica

CLAIRE and more



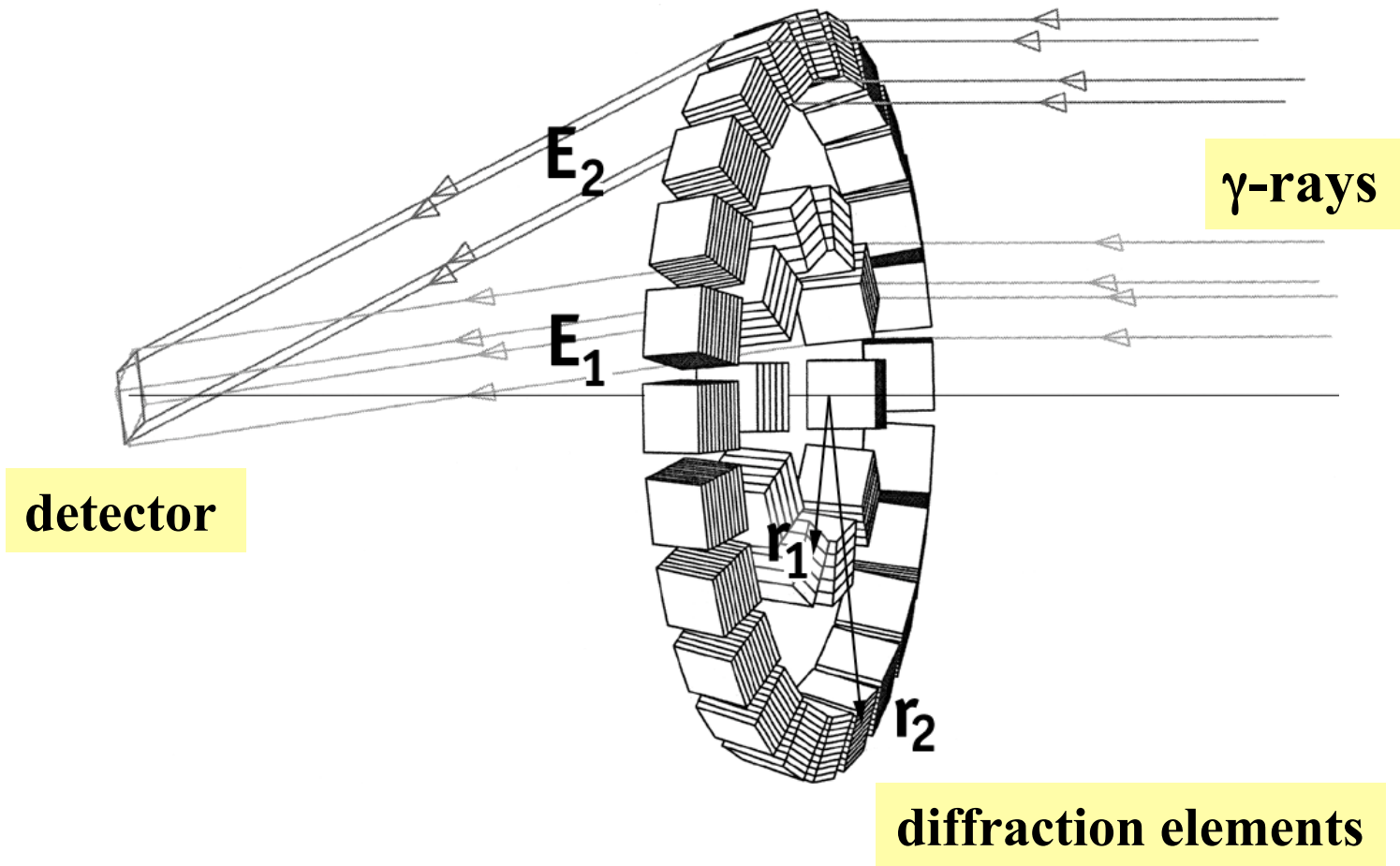
- P. von Ballmoos, H. Halloin, G.K. Skinner, P. Jean, J. Knödleseder, V. Lonjou, G. Vedrenne - **CESR, Toulouse, France**
- J. Evrard, Ph. Laporte - **CNES, Toulouse, France**
- P. Bastie - **Laboratoire de Spectrométrie Physique, Saint Martin d'Hères, France**
- B. Hamelin - **Institut Laue-Langevin, Grenoble, France**
- J. Alvarez, M. Hernanz, C. Badenes - **IEEC, Barcelona, Spain**
- R.K. Smither - **ANL, Argonne, USA**
- N.V. Abrosimov, A. Lüdge, H. Riemann, I. Rasin - **Institute for Crystal Growth, Berlin, Germany**
- V.N. Kurlov - **Institute of Solid State Physics of RAS, Chernogolovka, Russia**
- D. Borissova, V. Klemm - **Institute of Materials Science, Freiberg, Germany**
- A. Erko, A. Firsov – **BESSY, Berlin, Germany**
- O.V. Smirnova - **Soft-Impact Ltd., St. Petersburg, Russia**
- V.V. Kalaev, Yu.N. Makarov - **STR GmbH, Erlangen, Germany**

Phase diagram Si:Ge

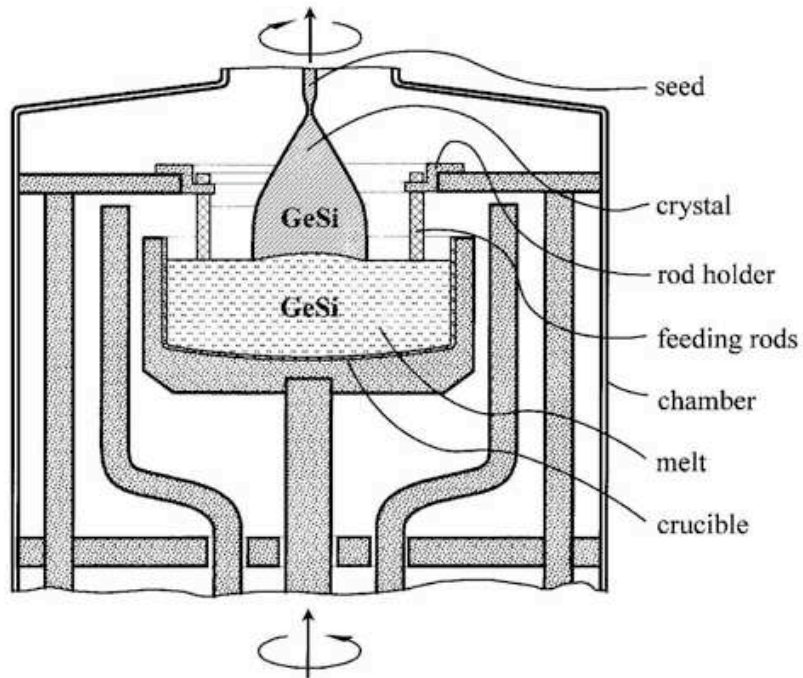




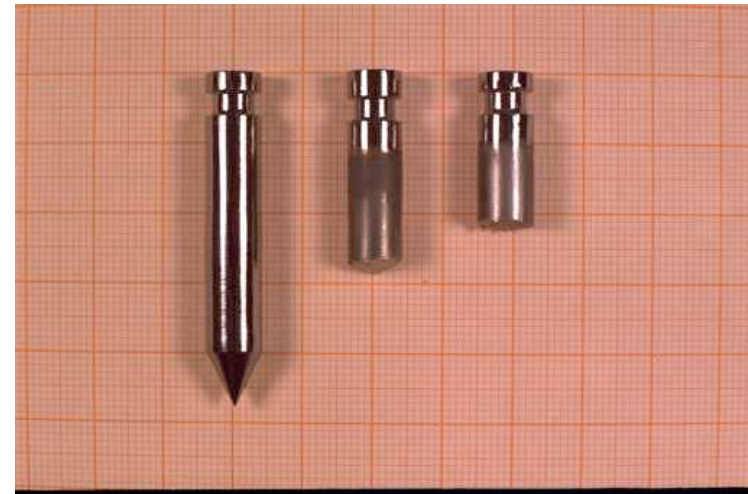
Focusing gamma-rays – the principles of Laue lenses



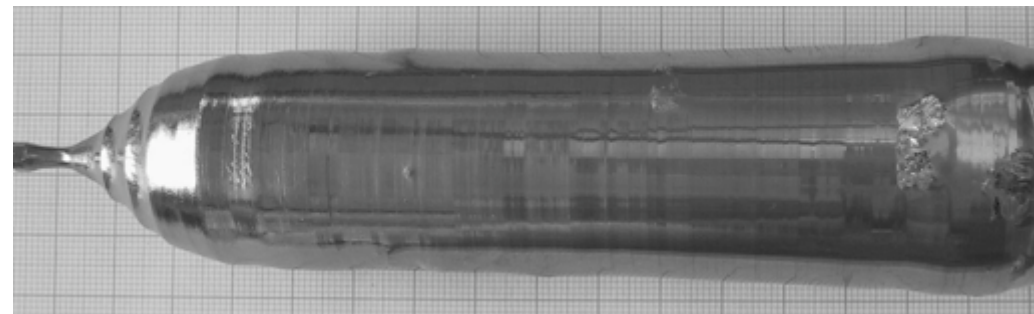
Czochralski growth of GeSi (mosaic) single crystals



feeding rods
before and after growth process



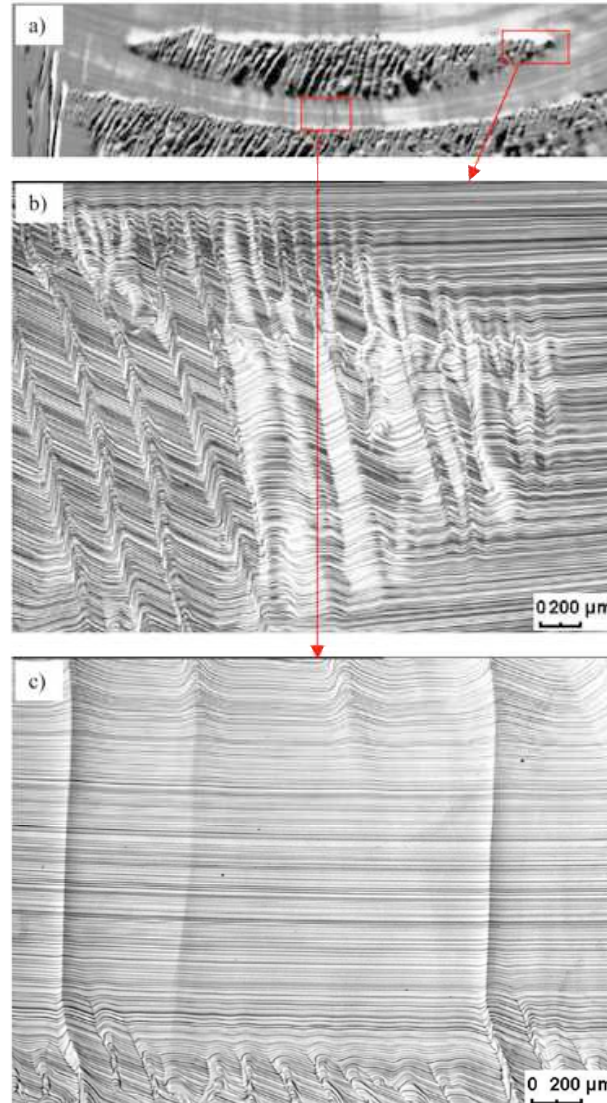
$\text{Ge}_{1-x}\text{Si}_x$
mosaic crystal
 $x = 2.0 \pm 0.15 \text{ at\%}$



Visualisation of the cellular structure



$\langle 112 \rangle$
growth direction



LPS measurement

38 x 10 mm

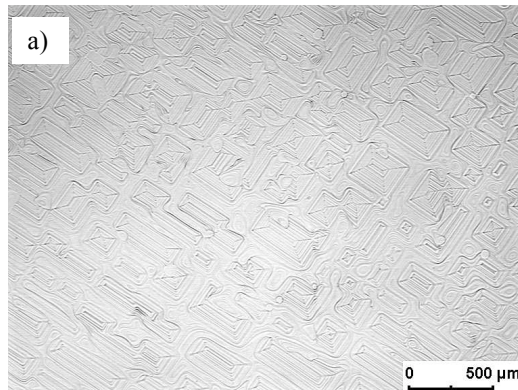
Etch pattern micrographs show the development of the cellular structure.

Features:

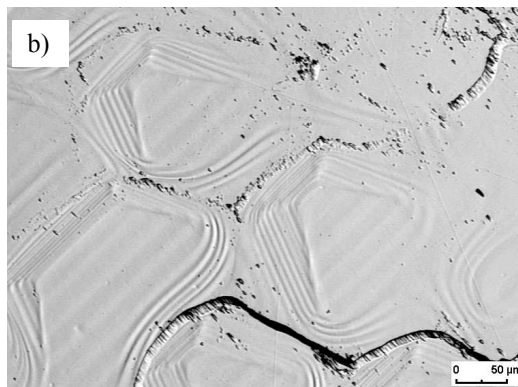
- the cell size increases during the growth
- the reversibility of the cellular structure (if only in some range)

longitudinal section
sample surface (110)

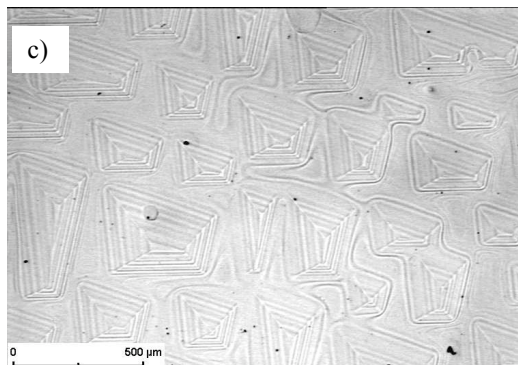
The cellular structure at the solid-liquid interface (etch patterns)



(100) growth direction
1.9 at% Si
mosaicity (not measured)



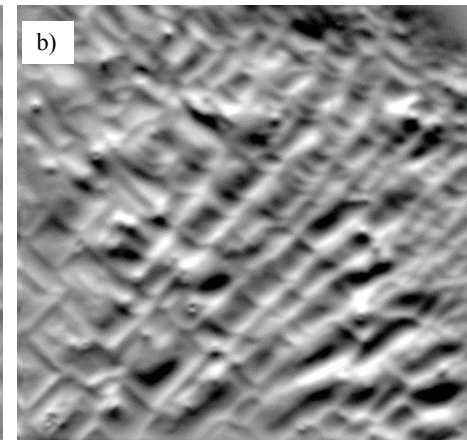
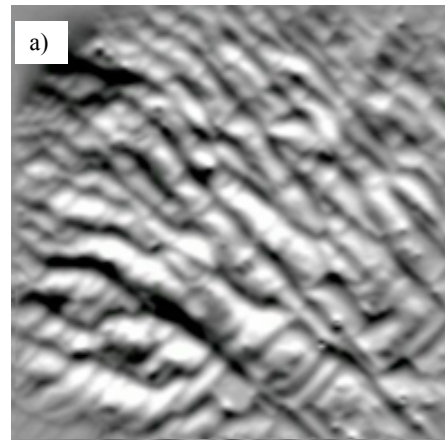
(112) growth direction
0.7 at% Si
mosaicity 20"



(130) growth direction
1.5 at% Si
mosaicity 42"



Cellular structure and diffraction properties

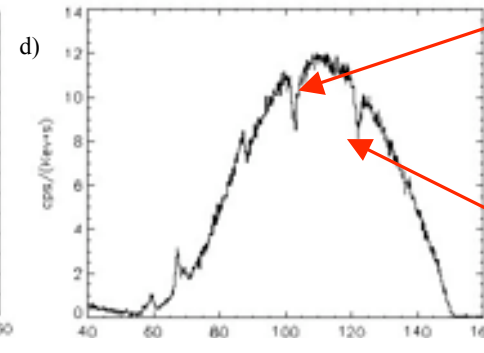
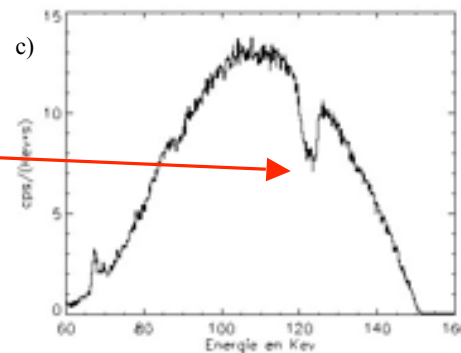


LPS measurement
10 mm x 10 mm
(100) surface orientation

transmitted beam

transmitted beam

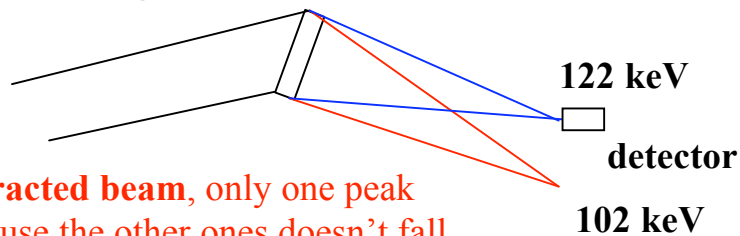
Centered at 122,4 keV
FWHM= 4,34 keV
Diffraction efficiency: 15%



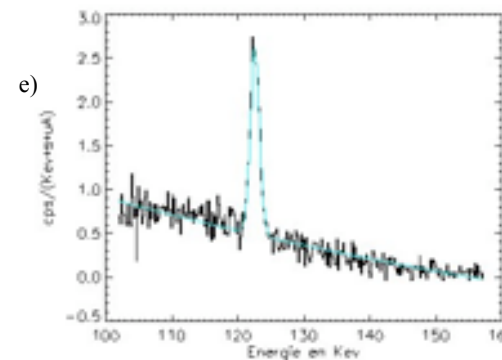
Peak 1 :
Centered at 102,9 keV
FWHM= 2,31 keV
Diffraction efficiency : 10%

Peak 2 :
Centered at 122,0 keV
FWHM= 1,82 keV
Diffraction efficiency: 8,5%

diffracting element



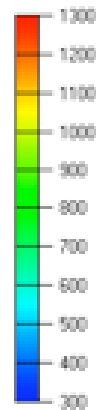
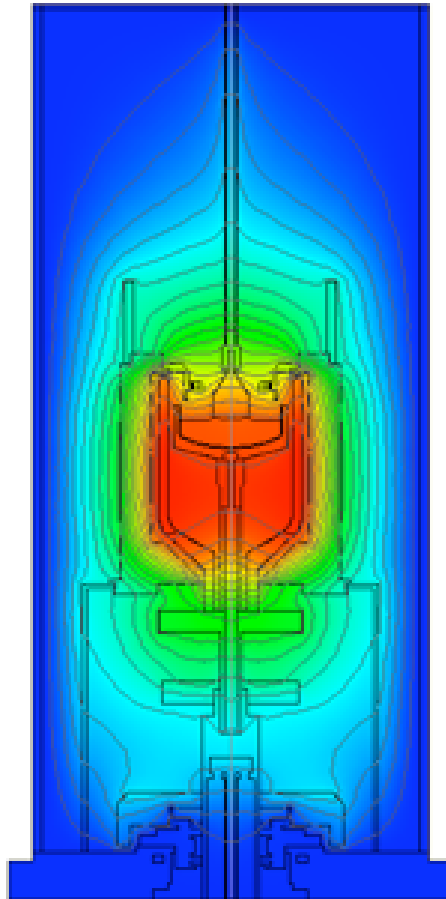
On the **diffracted beam**, only one peak targets because the other ones doesn't fall down on the detector for geometrical reasons



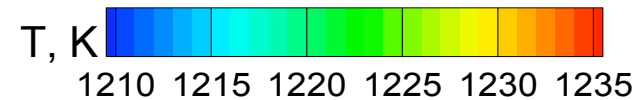
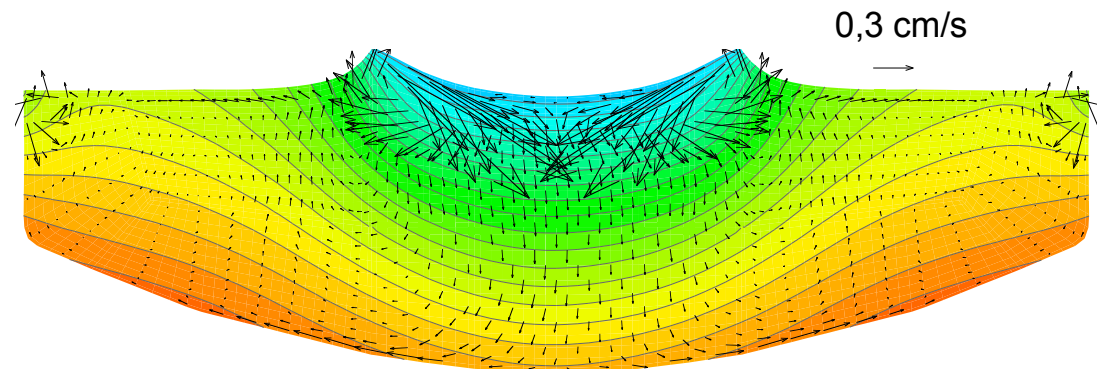
diffracted beam

Centered at 122,0 keV
FWHM= 1,62 keV

Global simulation of the growth process of GeSi crystal with mosaic structure

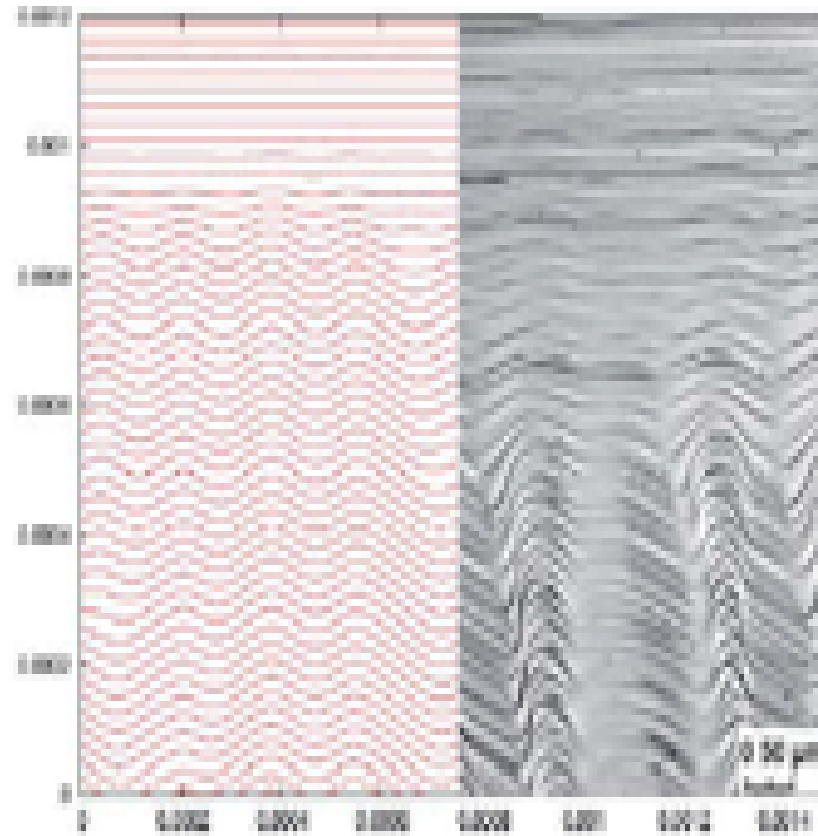


The temperature distribution in the whole growth setup obtained in 2D global heat computations



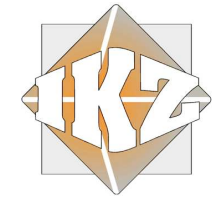
The averaged temperature distribution and the flow pattern in the melt obtained in the 3D computations during the growth of GeSi crystal

Simulation of growth interface of GeSi crystal with mosaic structure

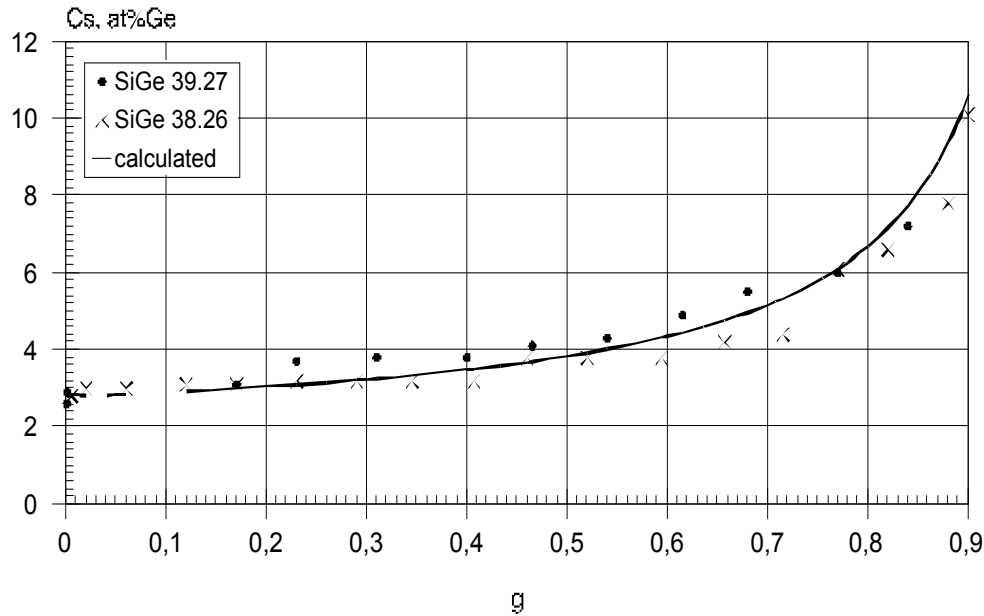


simulation

striations in GeSi crystal



Axial Ge distribution in Si_{1-x}Ge_x grown by Czochralski technique

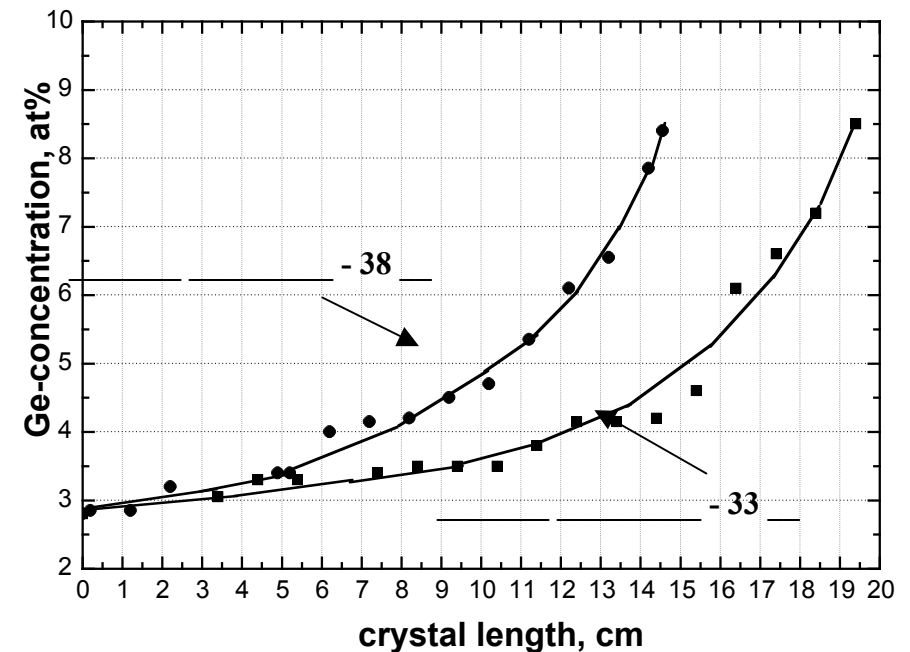


$$C_{Ge} = kC_{0,Ge}(1 - g)^{k-1}$$

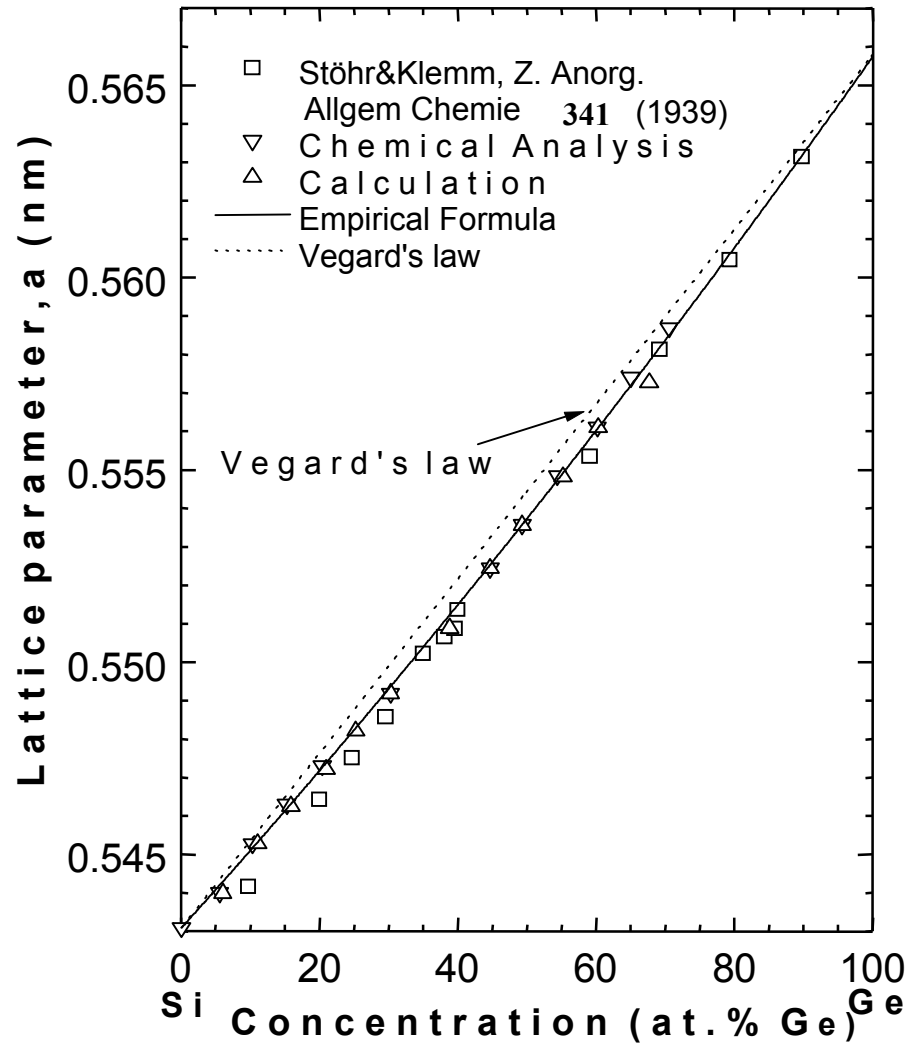
- C_{Ge} – Ge concentration in the crystal
- $C_{0,Ge}$ – initial Ge concentration in the melt
- k – distribution coefficient of Ge in Si
- g – solidified fraction

$$g = M_{cr} / M_0 = \pi \rho_{cr} R^2 L / M_0$$

- M_{cr} – current mass of the crystal
- M_0 – starting mass of the charge
- ρ_{cr} – crystal density
- S – crystal cross-section
- L – crystal length



Vegard's law



Vegard's law :

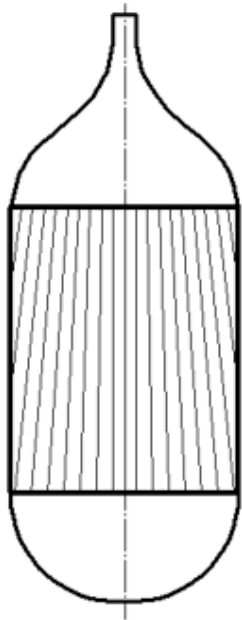
$$a_{\text{SiGe}} = x \cdot a_{\text{Ge}} + (1-x) \cdot a_{\text{Si}}$$

Empirical formula

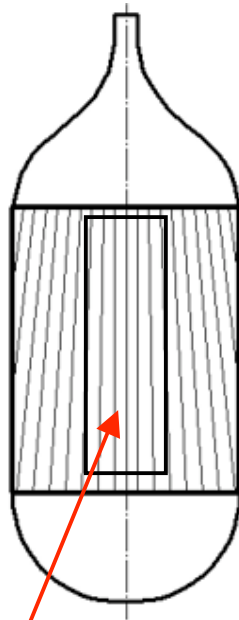
$$a_{\text{SiGe}} = (0.002733 \cdot x^2 + 0.01992 \cdot x + a_{\text{Si}}) \text{ nm}$$

$$a_{\text{Si}} = 0.5431 \text{ nm}$$

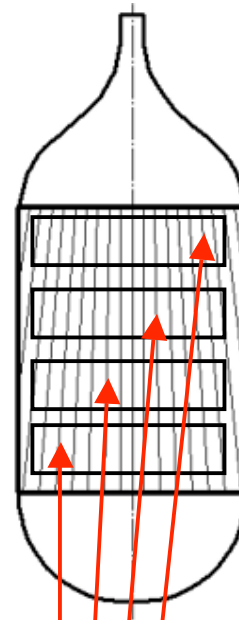
Some possibilities to cut monochromators from the gradient crystals



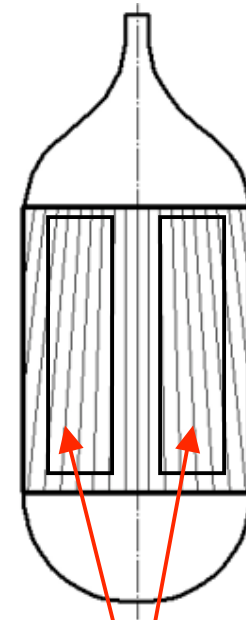
as-grown
gradient crystal



one non-banded
monochromator

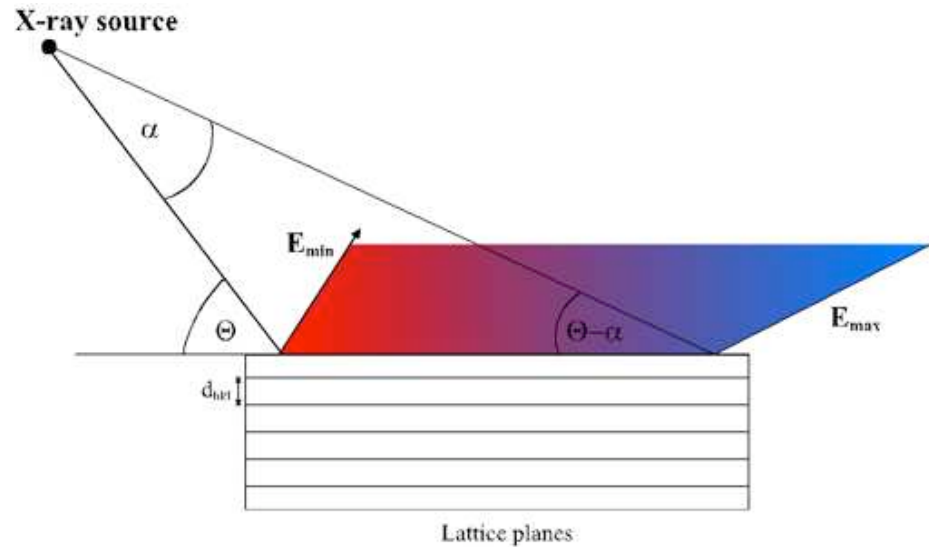


banded
monochromators



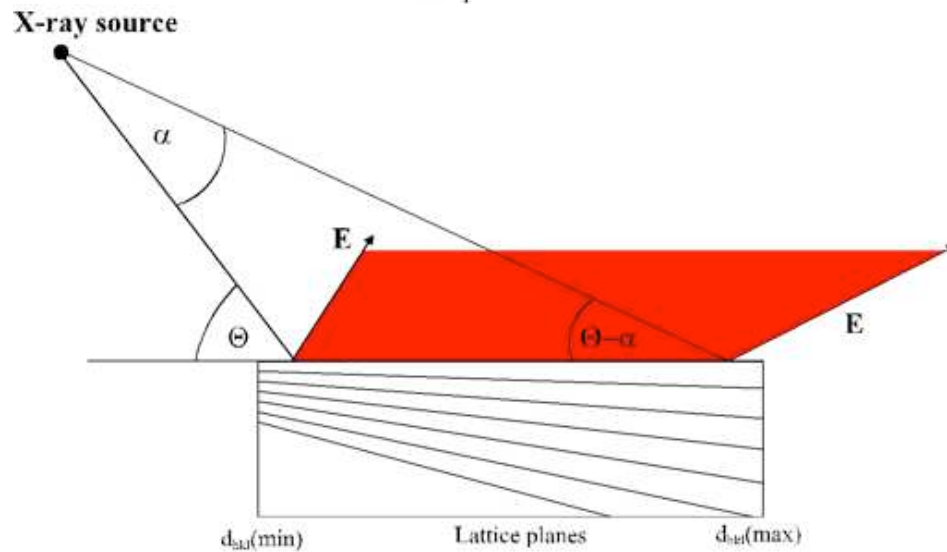
two banded
monochromators

SiGe gradient crystal as monochromator for X-rays



Bragg's law :

$$2 d \sin \Theta = \lambda$$



for gradient crystals

$$2 d (1 + \Delta d/d) \sin (\Theta - \alpha) = \lambda$$

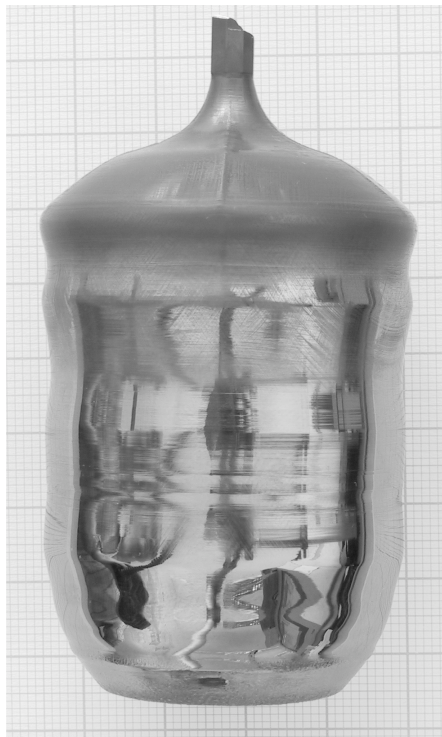
optimized gradient

$$B = (\Delta d/d)/L = \cos \Theta / R$$

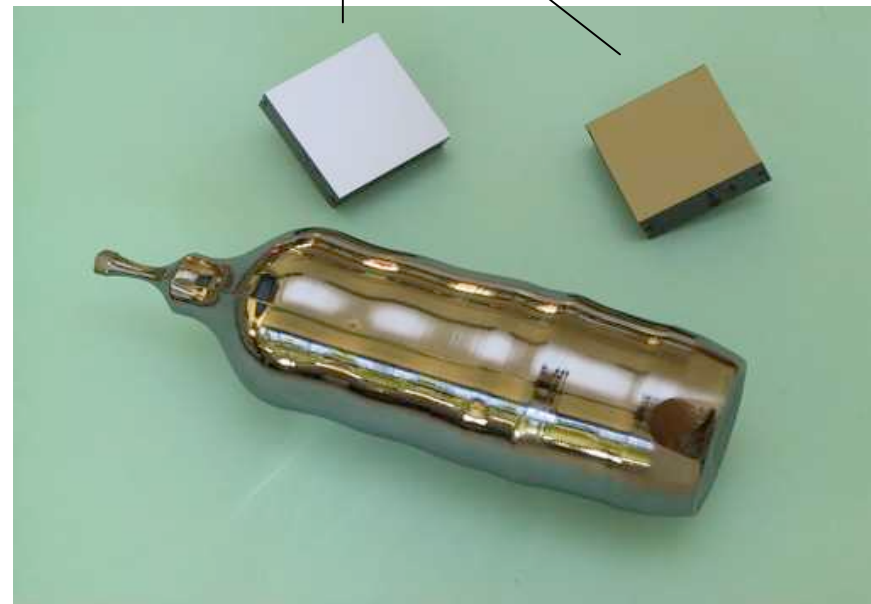
$\text{Si}_{1-x}\text{Ge}_x$ gradient crystals for high resolution synchrotron optics



↑ $\langle 112 \rangle$



SiGe-monochromators
 $30 \times 30 \times 10 \text{ mm}^3$

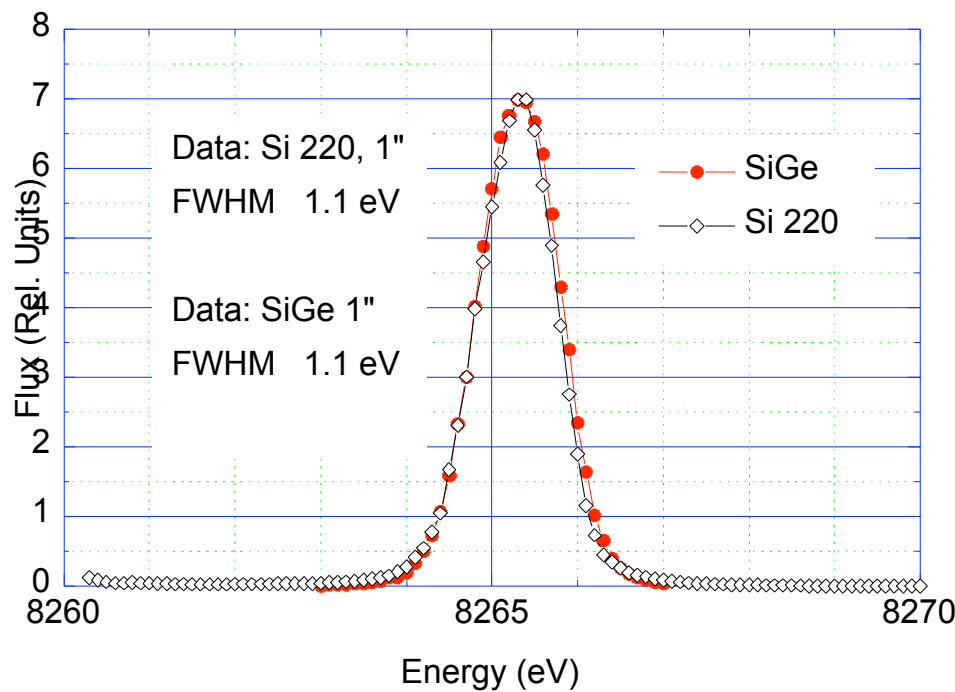


Energy spectrum from the SiGe (220) und Si (220) monochromators

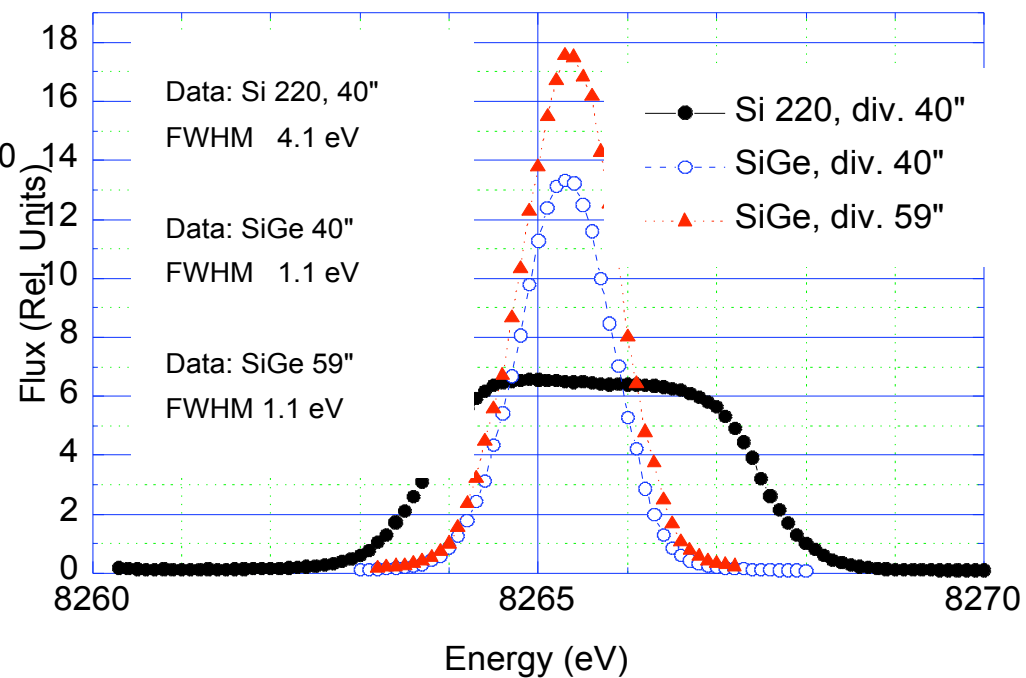


Reflection curves with a fixed Bragg angle

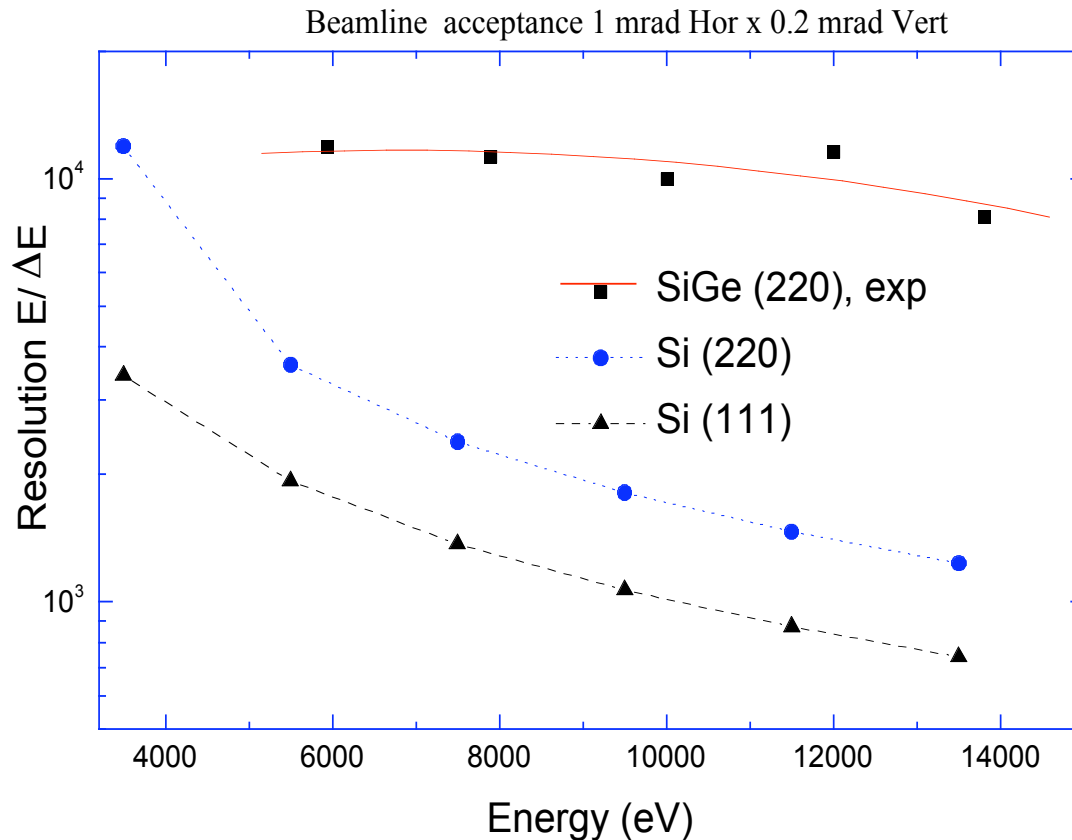
beam divergence 40 or 59 arc sec



beam divergence 1 arc sec



Energy resolution of the graded double SiGe (220) monochromator (in comparison with double Si (220) and Si (111) monochromators)



KMC-2 beamline (BESSY II)

Double crystal monochromator

2 SiGe (111) graded crystals

Energy range - 4 keV – 15 keV

Resolution - $E/\Delta E \sim 4000$

Exit flux - 10^7 - 10^{10} phot/sec/100mA

Spot size at experiment :

- 250 μm , horizontal

- 600 μm , vertical

- 5 μm x 5 μm (with capillary optics)

Instrumentation:

- EXAFS

- XANES

- X-ray diffractometry

- X-ray reflectometry



Calculation of crystal form to get constant gradient of Ge in $\text{Si}_{1-x}\text{Ge}_x$

$$C_{Ge} = kC_{0,Ge}(1 - g)^{k-1}$$

Axial distribution of impurities (Ge) in crystals grown by Czochralski technique

$$g = M_{cr} / M_0 = \pi\rho_{cr}R^2L / M_0$$

$$R = R(L)$$

$$\frac{\partial C_{Ge}}{\partial L} = B \Rightarrow \tilde{N}_{Ge} = BL + kC_{0,Ge}$$

$C_{0,Ge}$
 k

M_0

M_{cr}, ρ_{cr}, R, L

– initial Ge concentration in the melt

– distribution coefficient

– starting mass (charge)

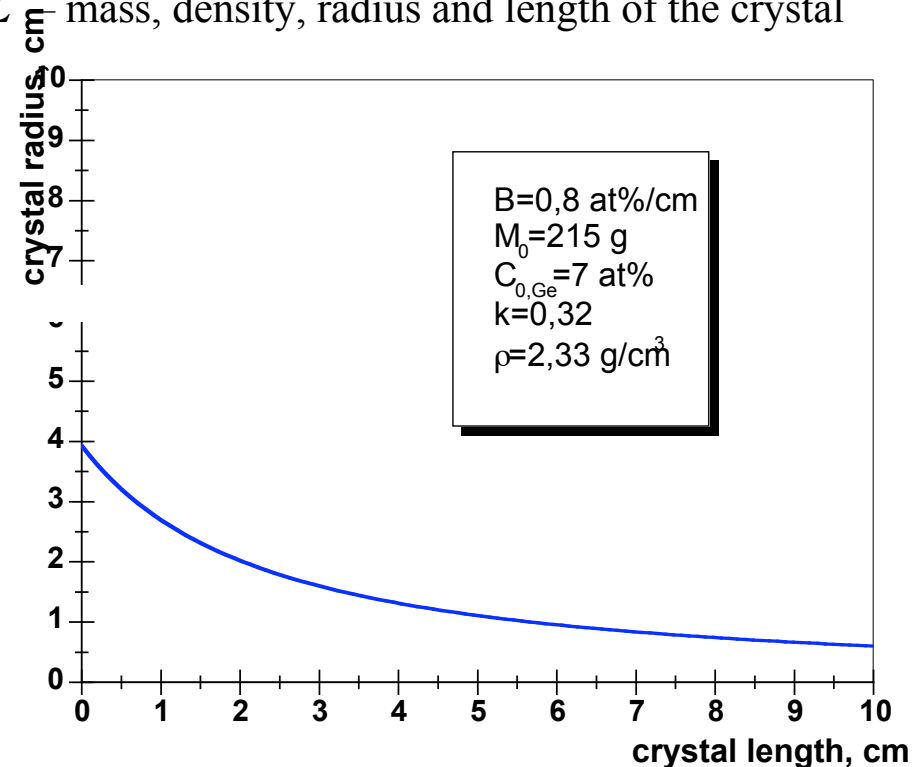
– mass, density, radius and length of the crystal

B –

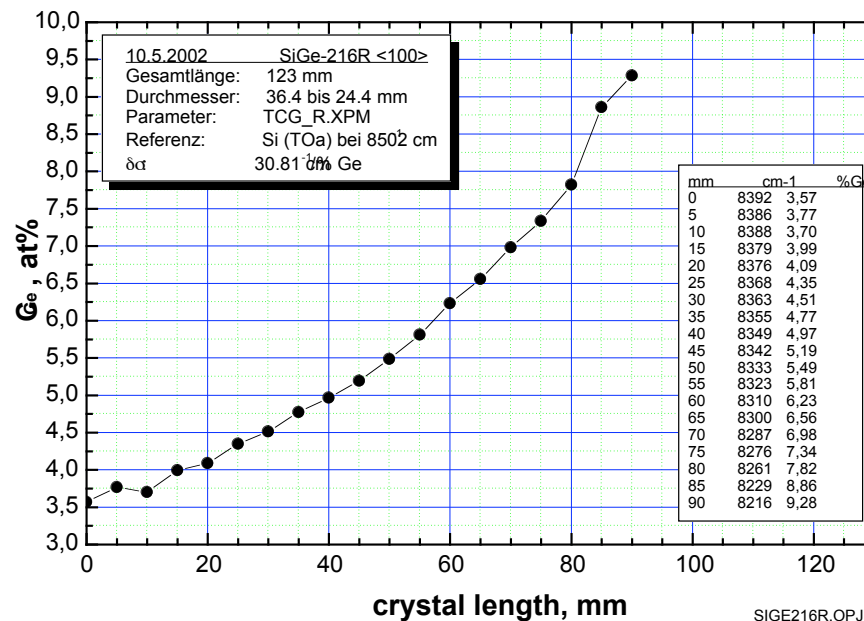
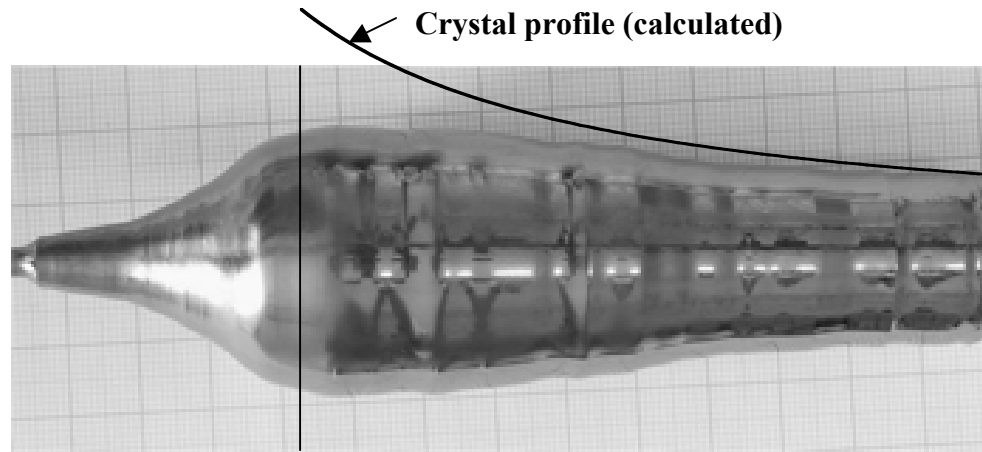
$$g = 1 - \left(\frac{B}{kC_{0,Ge}}L + 1 \right)^{\frac{1}{1-k}}$$

$$\frac{\partial g}{\partial L} = \frac{\pi\rho_{cr}}{M_0}R^2(L)$$

$$R(L) = \sqrt{\frac{M_0 B}{\pi\rho_{cr}(1-k)C_{0,Ge}} \left(\frac{B}{kC_{0,Ge}}L + 1 \right)^{\frac{2-k}{k-1}}}$$



Si_{1-x}Ge_x crystal with near constant gradient of Ge

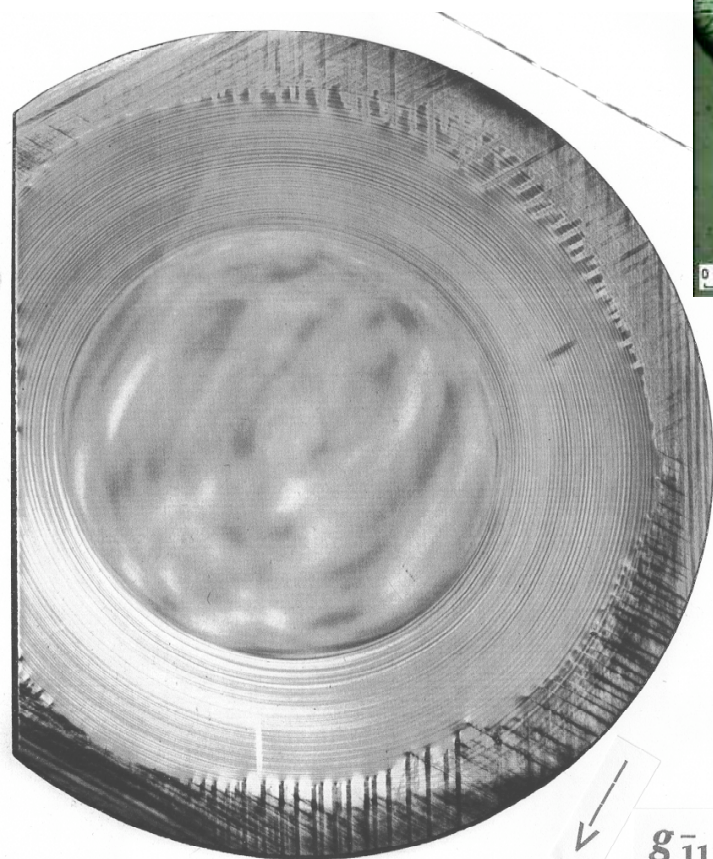
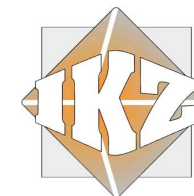


Conclusions



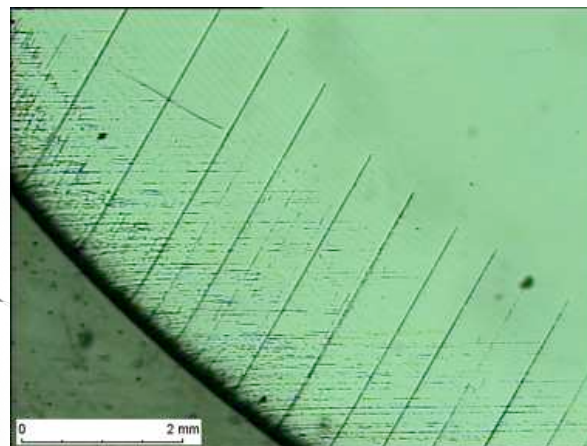
- A modified Czochralski technique could be used for the growth of $\text{Ge}_{1-x}\text{Si}_x$ mosaic crystals and $\text{Si}_{1-x}\text{Ge}_x$ gradient crystals. SiGe can be used for production of diffracting elements for gamma telescope lens.
- $\text{Ge}_{1-x}\text{Si}_x$ mosaic crystals were used to produce the diffracting elements for the CLAIRE gamma ray telescope. Although many of 556 diffracting elements had a diffraction efficiency up to 20 %, the overall efficiency of the lens is about 8.1 ± 0.7 % due to different diffraction properties of the elements
-

(111) SiGe-wafer with 4 at%Ge



X-ray topograph

g_{111}



Nomarski-contrast

